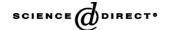


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Synthesis and photocatalytic activity of ZnO/ZnO₂ composite

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Abstract

 ZnO/ZnO_2 composite photocatalysts were synthesized by hydrothermal treatment at $120-180\,^{\circ}$ C of ZnO_2 , which in turn was obtained from an aqueous solution of $ZnSO_4$ and H_2O_2 . The composite particles showed morphology of ZnO prismatic crystallites with small ZnO_2 granules "fused" at surface. Photocatalytic activity was characterized based on photocatalytic degradation of methyl orange under UV-light (300 nm) illumination, and the maximum activity was observed for the powder with an intermediate ZnO_2 -to-ZnO conversion. The enhanced activity, as compared with the pure ZnO crystallites, has been attributed to the presence of the intimately bonded ZnO/ZnO_2 surface hetero-structure. This study also points out a new approach to synthesize a composite photocatalyst containing strongly coupled constituents by phase transformation reactions among the constituents.

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Keywords: Photocatalysis; ZnO; ZnO2; Hydrothermal synthesis

1. Introduction

ZnO is a wide-bandgap semiconductor having many applications, such as being a photocatalyst, transducer, varistor, phosphor, and transparent conductor [1], and wide varieties of synthesis techniques have thus been developed to produce this material in different forms [2]. They are, for instance, sol–gel process [3,4], hydrothermal process [5–7], chemical vapor deposition [8], and solution deposition [9,10]. It was particularly noticed that, in some of these processes [3,9,10], ZnO was obtained by phase transformation reaction from a pre-existing ZnO₂ matrix. Composites containing different proportions of these two phases can easily be obtained by varying the extent of the transformation. ZnO₂ is also a semiconductor with a bandgap greater than that of ZnO [9]. Nevertheless, the photocatalytic property of such a composite system has not been investigated.

Indeed, combination of more than one semiconductors has in some cases resulted in composite photocatalysts that are superior to any of its constituents in terms of photocatalytic activity [11–16], and it is in general believed that the properties at the interfacial regions between the constituents play a crucial role [11,12]. One key factor to such a synergic effect therefore, resides in intimate contact between the different phases. Conventionally, this was carried out either by calcining mixtures of the precursors of the constituent elements [13,14] or by depositing a second constituent phase onto the surface of the first, major one [15,16].

This work is thus motivated to explore the opportunity for enhanced photocatalytic activity in the ZnO/ZnO₂ system as compared with pure ZnO. In brief, ZnO₂ powder was first synthesized via a precipitation method and then subjected to hydrothermal treatment to form ZnO. Composites contain different ZnO/ZnO₂ proportions were obtained by hydrothermal treatments of ZnO₂ within the temperature range between 120 and 180 °C. Photocatalytic activity of the powders was characterized based on degradation of methyl orange [17]. The maximum activity was observed with an intermediate

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ZnO/ZnO₂ composition, which is associated with an intimately bonded ZnO/ZnO₂ surface hetero-structure.

2. Experimental

To synthesize ZnO₂ powder, 500 ml of $0.1\,\mathrm{M}$ ZnSO₄·7H₂O aqueous solution and 500 ml of 0.1 M NH₄OH aqueous solution were mixed to obtain a white zinc oxy-hydroxyl precipitate. After being washed with excessive de-ionized water, the precipitate was introduced into a 1 M H₂O₂ aqueous solution. The dispersion solution was then heated at 75 °C for 2h to result in ZnO₂ powder, which was subsequently washed with excessive de-ionized water and finally dried at 65 °C in air. The ZnO/ZnO₂ composite powders were synthesized by subjecting the ZnO₂ powder to hydrothermal treatment. In a typical hydrothermal run, the ZnO₂ powder was dispersed in de-ionized water in a stainless-steel autoclave with Teflon lining, and then heated at the selected temperature, ranging between 120 and 180 °C, for 2 h. After the hydrothermal treatment, the precipitate was collected, washed with de-ionized water and finally dried at 65 °C.

X-ray diffraction (XRD: Mac-Science/MXP3) was employed to determine crystalline phases and their (average) crystallite size based on the Scherrer equation. The conversion of the transformation reaction was determined based on the intensity ratio between the (102) reflection of ZnO and (220) reflection of ZnO2. These two reflections, although not the strongest reflection for either of the compounds, were chosen because they do not overlap with other peaks. For this purpose, a correlation curve of the XRD intensity ratio versus powder composition was constructed using mixtures of single-phased ZnO and ZnO2 powders prepared via the same processes as described above with known compositions. The powder morphology was analyzed by scanning electron microscope (Hitachi S-800). Diffuse reflectance ultraviolet-visible (UV-vis) spectroscopic analysis was conducted on a Hitachi U3410 spectrometer, which is equipped with an integration sphere and PbS and photomultiplier detectors for photon detection. Deuterium and tungsten iodide lamps are used as the light sources for wavelength ranges of 185-340 and 340-2500 nm, respectively. The spectra were acquired at room temperature with BaSO₄ as the reference.

Kinetic experiments were carried out by using a vertical tubular batch reactor made of quartz. During a typical run, the entire reactor was enclosed inside a UV-light house (Rayonet photochemical reactor, RPR-100), and the reactor was partially filled with an aqueous methyl orange solution, in which the oxide particles, in an amount of 0.25 g/l, were constantly dispersed by a magnetic stirrer. The light house was equipped with eight UV-light tubes (maximum intensity at 300 nm, Fig. 1), each having a power of 12 W. The powder was dispersed in the solution for 60 min in dark, in order to reach the adsorption equilibrium of methyl orange on the

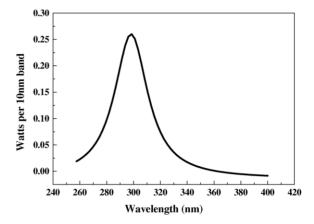


Fig. 1. Energy distribution of the UV-light source employed in the kinetic study.

powder, before UV-light was turned on. Small amounts of solution were retreated at the very beginning as well as intermittently during the course of reaction. The methyl orange concentration was determined by measuring the absorption at $\lambda = 465$ nm, on a UV-vis spectroscopy (GENESYSTM 10 SERIES). A correlation curve between methyl orange concentration and the absorption was pre-established.

3. Results and discussion

XRD showed that the powder produced by $\rm H_2O_2$ at 75 °C is single-phased $\rm ZnO_2$ having an Scherrer crystallite size of 5.4 nm (curve 1, Fig. 2). Upon hydrothermal treatment, the powder remained single-phased $\rm ZnO_2$ at 120 °C, while $\rm ZnO$ started to appear at 130 °C. With further increasing temperature, the amount of $\rm ZnO$ continued to grow at the expense of $\rm ZnO_2$ (Fig. 2). No any other phase was detected during the course of transformation, suggesting a simple one-step transformation mechanism:

$$ZnO_2 \rightarrow ZnO_{(s)} + \frac{1}{2}O_{2(g)}$$
.

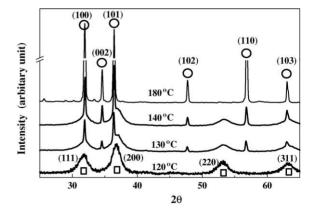


Fig. 2. XRD patterns of the powders synthesized at different temperatures. Key: (\bigcirc) reflections of ZnO; (\Box) reflections of ZnO₂.

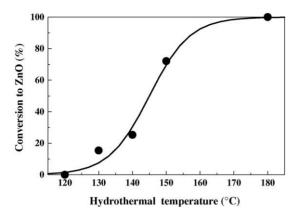


Fig. 3. Phase transformation conversion from ZnO_2 to ZnO vs. hydrothermal temperature.

The conversion data (Fig. 3) indicate that the transformation accelerates at temperatures exceeding 140 °C, and the powder became single-phased ZnO after treatment at 180 °C. It was also noticed that the reflections of ZnO are very sharp, indicating large crystallites, even for the powders of low conversion, such as the 130 and 140 °C samples. Calculation

based on the Scherrer equation indicated average ZnO crystallite sizes exceeding 40 nm in all samples (exact values cannot be determined with high confidence because large error can easily arise from choosing the baseline of the XRD spectra due to the sharpness of the peaks).

SEM analysis showed that the ZnO₂ starting powder has granular morphology with a rather uniform particle size of \sim 0.25 μ m (Fig. 4a). On the other hand, the fully transformed, 180 °C sample showed rod-like ZnO prisms having a length from \sim 3 to 5 μ m and \sim 1 μ m in thickness (Fig. 4b). The 140 °C powder, which has a conversion of 25%, shows a small number of large ZnO prisms (Fig. 4c) having a dimension of a few microns instead of large amount of small ZnO crystallites. This is consistent with the XRD results, which show very sharp ZnO peaks even at low conversion. These results may suggest that nucleation be the rate-limiting step for the transformation, and therefore, stable nuclei grow rapidly while the number of nuclei remains small. In addition, it was noticed that the surfaces of the ZnO crystallites in the partially transformed powders were mostly "fused" with ZnO2 granules (Fig. 4c). The surface density of the granules is abundant for the 140 °C crystallites, but significantly decreases

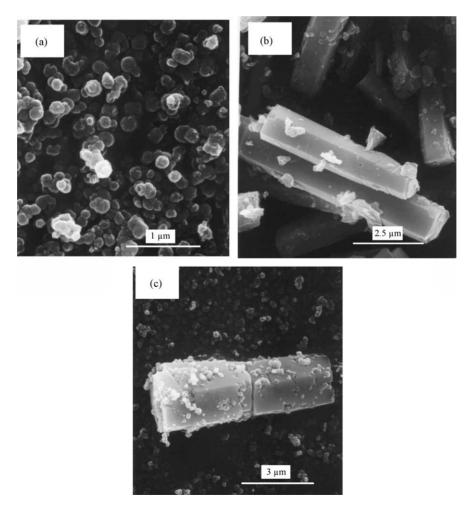


Fig. 4. SEM micrographs: (a) ZnO₂; (b) ZnO obtained at 180 °C and (c) ZnO/ZnO₂ at 140 °C.

as the transformation approaches completion at 180 °C (Fig. 4b).

This is the first time ZnO₂-to-ZnO transformation has been carried out under hydrothermal condition. Uekawa et al. [4] prepared ZnO₂ by using the same procedures as in the present study but carried out the transformation reaction by firing ZnO₂ under dry condition (the atmosphere was not explicitly specified but most likely was either air or oxygen). The resulted powders were ZnO nanoparticles of 19 and 41 nm after firing at 300 and 500 °C, respectively. Heat treatment of highly crystallized ZnO₂ thin film in air or vacuum [9,10] also resulted in nanocrystalline ZnO. These results are in great contrast with the µm-sized ZnO crystallites obtained here. The difference may indicate that the nucleation step is greatly accelerated and no longer be rate limiting under their synthesis conditions. Thus, the ZnO crystallites containing surface ZnO₂-granules are unique to the hydrothermally effected ZnO2-to-ZnO transformation. Direct hydrothermal synthesis of ZnO from Zn oxy-hydroxide precipitate does not create similar morphology [5–7].

The UV-vis spectra have been expressed in Kubelka-Munk unit (K/S), as shown in Fig. 5, where K is the absorption coefficient, and S, the scattering coefficient. The starting ZnO₂ powder showed an absorption edge at 300 nm, corresponding to a bandgap of 4.12 eV. The bandgap is smaller, by nearly 0.1 V, than that reported for a highly crystallized ZnO₂ thin film prepared by a solution deposition method [10], and the difference may be because the ZnO2 powder synthesized here has poorer crystallinity. This is also indicated by the broad transition in absorption near the edge energy. When ZnO started to appear after hydrothermal treatment at 130 °C, the spectrum showed a two-step profile with an additional absorption at 395 nm. With increasing hydrothermal temperature, the ZnO₂-absorption decreased, while the ZnOabsorption increased and shifted gradually to higher wavelengths. The profile finally showed single-step ZnO absorption with the absorption edge at 395 nm (3.13 eV) for the 180 °C sample. These results are consistent with the XRD data in terms of the phase composition.

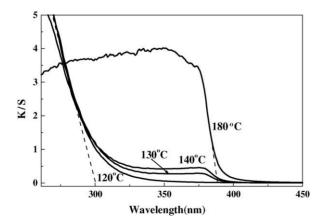


Fig. 5. UV-vis spectra of the powders synthesized at different temperatures.

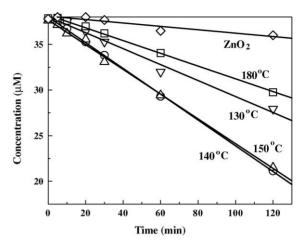


Fig. 6. Concentration of methyl orange vs. time under irradiation of UV-light.

The kinetic study was carried out by monitoring the absorption of the solution at 465 nm wavelength. A decrease in this absorption corresponds to degradation of polyaromatic rings in methyl orange to create mono substituted aromatics [17]. Fig. 6 shows methyl orange concentration versus time under irradiation for different powders. The ZnO₂ powder, either fresh prepared or hydrothermally treated at 120 °C, exhibited very low activity. This is rather expected, because the absorption edge wavelength (300 nm) of the ZnO₂ powder is the same as the maximum-intensity wavelength of the light source, and the efficiency for generating electrons and holes is hence expected to be very low. This may additionally be compounded by the poor crystallinity of the powder, which shortens the lifetimes of the excitons. For powders that were synthesized at 130 °C or above, the methyl orange concentration was found to decrease almost linearly with time. The linearity suggests the zero-order reaction kinetics, and the slope is equal to the rate constant. The rate constant (Fig. 7) was found to first increase and then decrease with increasing ZnO content, showing a maximum for the powder synthesized within 140–150 °C. The presence of an optimum activity at an intermediate ZnO/ZnO₂ ratio may

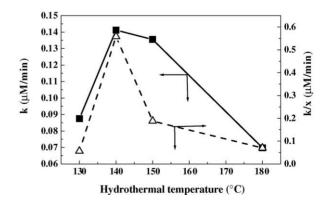


Fig. 7. Rate constant (k) and normalized rate constant (k/x) vs. hydrothermal temperature, where x is the ZnO₂-to-ZnO conversion, for photocatalyzed degradation of methyl orange.

indicate an important role played by the interface of the two phases.

The physical and chemical mechanisms involved in semiconductor photocatalysis have been reviewed in detail in the literature (e.g., [11,12]). In brief, the processes involved can roughly be divided into three groups. The first group includes the physical processes taking place within the solid catalyst to result in a supply of charge carriers to the catalyst surface. They are photo-excited generation of electron in conduction band and hole in valence band, and diffusion and recombination of the generated charge carriers. The second group includes a series of surface reaction steps leading to chargetransfer at the solid–fluid interface. And the third group includes the transport processes of the chemical species (reactants/products) on the fluid side of the interface. As these three groups operate in series, the overall reaction kinetics is limited by the group that is the slowest, i.e., rate-limiting.

In the case where either the interfacial reaction or transport of chemical species on the fluid side is rate-limiting, the overall reaction rate would show non-zero reaction order. On the other hand, if the reaction rate is limited by the supply of the charge carriers on the solid side to the interface, one expects the reaction rate to exhibit zero-order concentration dependence. This is the case observed for the methyl orange degradation in the present study (Fig. 6). That is, for all the samples shown here, the overall reaction activity toward methyl orange degradation is determined by the total arriving rates of charge carriers at the catalyst surface.

As shown in Figs. 6 and 7, the photocatalytic activity does not show monotonic correlation with the hydrothermal temperature, i.e., with the amount of ZnO. Rather, some of the partially transformed powders, particularly the 140 and 150 °C powders, showed higher activities than the singlephased ZnO powder synthesized at 180 °C. To understand these differences, two factors are considered, including the total active surface area and the arrival flux (per unit time and area) of charge carriers to the catalyst surface. Because of the wavelength of the light source employed, we may consider that only the surface area of ZnO is of primary importance. As described early, the transformation reaction appeared to be nucleation-limited, and XRD and SEM results showed that the ZnO phase in different powders possess similar grain and particle sizes. As a result, the specific surface area of the ZnO phase should be similar in all powders, and therefore, the total active surface area should increase with increasing amount of ZnO. Thus, the non-monotonic trend in catalytic activity is not likely due to the variation in active surface area.

Assuming similar specific surface areas of ZnO in these powders, the value of the reaction constant divided by transformation conversion (right *y*-axis, Fig. 7), referred as the normalized rate constant, then measures the specific surface activity, and in this case the arrival flux of charge carriers to the surface. The first rise in specific surface activity observed for hydrothermal temperature between 130 and 140 °C may be due to improved crystallinity of the synthesized ZnO phase with an increase in hydrothermal temperature. On the

other hand, at higher hydrothermal temperatures, the specific surface activity are in the order of 140>150>180 °C. This trend was found to be correlated with the amount of the ZnO₂ granules present on the ZnO crystallites. That is, a higher specific surface activity corresponds to a greater amount of ZnO₂ surface granules. It is suggested that the ZnO/ZnO₂ interface may play an important role in enhancing the arrival flux of charge carrier to the ZnO surface. One possible effect due to the presence of the ZnO2 granules is that the difference in band structure between ZnO and ZnO2 could result in potential gradient at the interface, which facilitates separation and hence reduces recombination between the excited holes and electrons. In addition, the structural defects created near the ZnO/ZnO2 interfaces could also serve as favorable trap sites of the excitons to reduce their recombination. In either case, a higher flux of charge carrier resulted from a longer carrier life time. Detailed photo-absorption and cathodoluminescence spectroscopic analyses on individual ZnO crystallites are underway.

4. Conclusion

 $\rm ZnO/ZnO_2$ composite photocatalyst containing $\rm ZnO$ crystallites bonded with surface $\rm ZnO_2$ granules have been synthesized by a hydrothermal treatment of $\rm ZnO_2$. The composite has shown a superior photocatalytic activity as compared with either of its constituents toward degradation of methyl orange. The enhancement has been ascribed to the presence of either structural or energetic heterogeneity within the interfacial regions between the two oxides. This study has also pointed out a new approach to synthesize a composite photocatalyst containing strongly coupled constituents, namely via one or multiple steps of transformation reaction among the constituents.

Acknowledgment

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